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Ultrathin polymer membranes with patterned, micrometric pores for organs-on-chips

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9 2 micrometric pores for organs-on-chips
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44 16 **KEYWORDS.** Semipermeable ultrathin polymer membranes, microporous ultrathin polymer
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46 17 films, spin coating, microneedles, femtosecond laser machining, polymer replication.
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48 18 **ABSTRACT**
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51 19 The *basal lamina* or basement membrane (BM) is a key physiological system that
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53 20 participates in physicochemical signaling between tissue types. Its formation and function are
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55 21 essential in tissue maintenance, growth, angiogenesis, disease progression, and immunology. *In*
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3 22 *in vitro* models of the BM, *e.g.*, Boyden and transwell chambers, are common in cell biology and
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5 23 lab-on-a-chip devices where cells require apical and basolateral polarization. Extravasation,
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7 24 intravasation, membrane transport of chemokines, cytokines, chemotaxis of cells, and other key
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9 25 functions are routinely studied in these models. The goal of the present study was to integrate a
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11 26 semipermeable ultrathin polymer membrane with precisely positioned pores of 2 μm diameter in
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13 27 a microfluidic device with apical and basolateral chambers. We selected poly (L-lactic acid)
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15 28 (PLLA), a transparent biocompatible polymer, to prepare the semipermeable ultrathin
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17 29 membranes. The pores were generated by pattern transfer using a three-step method coupling
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19 30 femtosecond laser machining, polymer replication, and spin coating. Each step of the fabrication
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21 31 process was characterized by scanning electron microscopy to investigate reliability of the
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23 32 process and fidelity of pattern transfer.

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29 33 In order to evaluate the compatibility of the fabrication method with organs-on-a-chip
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31 34 technology, porous PLLA membranes were embedded in polydimethylsiloxane (PDMS)
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33 35 microfluidic devices and used to grow human umbilical vein endothelial cells (HUVECS) on top
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35 36 of the membrane with perfusion through the basolateral chamber. Viability of cells, optical
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37 37 transparency of membranes and strong adhesion of PLLA to PDMS were observed, thus
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39 38 confirming the suitability of the prepared membranes for use in organs-on-a-chip devices.

39 INTRODUCTION

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41 40 *In vitro* models are an essential part of cell biology. Transwells have been widely adopted
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43 41 for polarized cell and migration assays since Boyden's initial chemotaxis experiment.¹ The
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45 42 promise of organs-on-a-chip technology is essentially to create a more robust model of the
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47 43 complex electrophysicochemical systems that control cell function and fate.^{2,3} In static transwells
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49 44 as well as complex microfluidic organ models, a key element is the basal lamina or basement
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3 45 membrane (BM).⁴ This membrane supports diverse epithelial cell types and participates in
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5 46 signaling and transport for the system.⁵ In contrast to the static transwell and Boyden chamber
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8 47 devices, microfluidic barrier devices offer the possibility of using physiologically realistic flow
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10 48 of cell culture media so that the resulting shear stresses polarize the endothelial cells on the
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12 49 vascular side of the BM, as might be found in organ-on-chip models of the blood-brain barrier.^{6,7}
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15 50 Critical properties of a semipermeable membrane in a microfluidic device include controlled
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17 51 porosity, high species flux, mechanical strength, surface biocompatibility, and optical
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20 52 transparency. Polycarbonate and polyester membranes are the main materials for transwell cell
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22 53 culture and diffusion experiments. These membranes are far thicker than the BM and the
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24 54 transparency is not optimal in bright field light, confounding observation by differential
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27 55 interference contrast. Ultrathin polymer films (nanofilms) are promising candidates for the
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29 56 membrane model system. They are closer in thickness to native BM, are optically transparent
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31 57 and compatible with cell culture.

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34 58 Nanofilms with thicknesses ranging from tens to hundreds of nanometers belong to an
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36 59 interesting class of polymeric nanomaterials having lateral-dimension-to-thickness ratios that are
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39 60 greater than 10^6 , and unique thickness-dependent interfacial and mechanical properties. In
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41 61 particular, previous studies have revealed that such films can have non-covalent high
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43 62 adhesiveness to surfaces, tunable flexibility and molecular permeability,⁸ defined structural
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45 63 color,⁹ and mechanical strength,¹⁰ and possibly conductive¹¹ and magnetic properties.¹² These
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48 64 characteristics make such films a promising solution for multiple applications in the biomedical
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50 65 field. As an example, thanks to their quasi-two-dimensional nature, such nanofilms resemble the
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53 66 *lamina basalis* of the extracellular matrix in human tissues, the ideal structure to direct the
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55 67 cellular organization and function thereby enabling organ regeneration and function.

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4 68 Nanofilm fabrication techniques are diverse, and include layer-by-layer,¹³ Langmuir–
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6 69 Blodgett methods¹⁴ and others. The simplest route to fabricate freestanding and easy-to-handle
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8 70 nanofilms is spin coating from a liquid polymer-solvent solution.¹⁵ In previous works, full
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10 71 characterization of plain nanofilm structures was performed. The adhesion and proliferation of
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12 72 different cell types on poly (lactic acid) (PLLA) nanofilms¹⁶ and polyelectrolytic films¹⁷ was
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14 73 confirmed. We now report on the fabrication and characterization of ultrathin semipermeable
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16 74 PLLA membranes for organs-on-a-chip applications. Herein “semipermeable membranes” refer
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18 75 to membranes with ordered pores of a specific size, distribution and shape that are precisely
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20 76 patterned to allow passage of chemical species and cells of a specific size range, while blocking
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22 77 migration of other cells. The semipermeable membranes function as support for growth of cell
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24 78 layers and as separation between distinct microfluidic chambers.

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29 79 Existing examples of organs-on-a-chip have used perforated polydimethylsiloxane
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31 80 (PDMS) membranes,^{18,19} polycarbonate^{7,20} or photoresist membranes.^{21,22,23} While the PDMS
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33 81 membranes have the advantages of optical transparency, tunable elastic modulus and can be
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35 82 easily integrated in a monolithic PDMS device, it is still difficult to reduce the thickness of the
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37 83 membrane below 5 μm . Submicron membrane thickness is essential for physical contact and
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39 84 paracrine communication between cells growing on both sides.

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43 85 The engineered semipermeable ultrathin polymer membranes are an advancement over
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45 86 current solutions. Numerous approaches have been used to fabricate organic semipermeable
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47 87 membranes,²⁴ such as polymer synthesis²⁵ and ion track etching of polymers.²⁶ However, most of
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49 88 these methods have drawbacks, mainly related to random arrangement of pores with a wide size
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51 89 distribution or nanometric/sub-micrometric pore size. Other methods use e-beam lithography and
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53 90 by focused ion beam milling for direct micro- and nanofabrication of the pores on an otherwise
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3 91 unpatterned polymer sheet.^{27,28} These approaches have the advantage of control on the design
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5 92 and arrangement of the pores,²⁹ but they also present drawbacks, such the length of time and
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8 93 hence cost required for serial, pore-by-pore fabrication of each membrane. Moreover, while
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10 94 direct etching of any polymeric film should be theoretically possible for example by tuning
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12 95 current and voltage of an ion or electron beam,²⁹ it is practically impossible to drill holes with
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14 96 micrometric diameter through a ~100 nm thin film. The Gaussian distribution of the ion beam
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16 97 and the control of current and milling depth are not sufficient to avoid local polymer melting.
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18 98 Debris from the milling process limits the depth to 10x the diameter of the hole. Furthermore,
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20 99 direct milling of the film would implicate local changes in the film mechanical and structural
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22 100 characteristics and thus increase the risk of breakage and tears in the polymer film.
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27 101 Alternatively, ultrathin polymer membranes containing precisely patterned micrometric
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29 102 pores can be formed by spin coating assisted deposition of a polymer-solvent solution atop a
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31 103 sacrificial array of spatially ordered polyvinyl-alcohol (PVA) nanoneedles, followed by removal
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33 104 of the sacrificial PVA nanoneedle array. The PVA nanoneedles are formed by PVA replication
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35 105 of an array of surface nanometric pores formed on a fused silica wafer using femtosecond laser
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37 106 machining. The details and suitability of this novel approach are outlined herein for PLLA
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39 107 membranes.
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109 **MATERIALS AND METHODS**

110 **Fabrication of fused silica molds**

111 As a first step towards creating the PVA needle array, a 10 mm × 10 mm regular array of
112 201×201 pores was patterned on the surface of a 500 μm thick UV grade fused silica wafer
113 (Mark Optics, Inc.) using the single-pulse femtosecond laser machining technique first described

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3 114 by White, et al.³⁰ In their 2008 paper, White and co-workers demonstrated that high numerical
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5 115 aperture single-pulse femtosecond laser machining can create uniquely-shaped pores at the
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8 116 surface of fused silica, exhibiting high aspect ratios, with depths that exceed 10 μm and
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10 117 diameters below 200 nm. In the present work, femtosecond laser machining was carried out
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12 118 using the system described by Rajput.³¹ A dry microscope objective, namely a Nikon CF Plan
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14 119 Achromat 79173, was used to focus the femtosecond laser beam on the surface of the fused silica
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16 120 wafer. Energy per pulse of 3.2 μJ was used to open each pore. The laser-patterned wafer was
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18 121 soaked first in a 5 M KOH solution at 80°C for 2 hours, and then in de-ionized water also at
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20 122 80°C for another 2 hours, in order to remove any machining debris. The washed wafer was dried
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22 123 under a stream of nitrogen, and used to prepare PLLA membranes for characterization purposes.
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27 124 A second fused silica mold, designed to prepare PLLA membranes for use in organs-on-
28
29 125 a-chip trials was femtosecond laser machined in a similar way. This mold was patterned with a
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31 126 16 row x 100 column rectangular array of surface pores. The distance between rows was 400 μm ,
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33 127 while the distance between columns was 100 μm , thereby creating pores whose arrangement
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35 128 matched channels in the microfluidic device.
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39 129 **Preparation of PVA nanoneedle arrays**

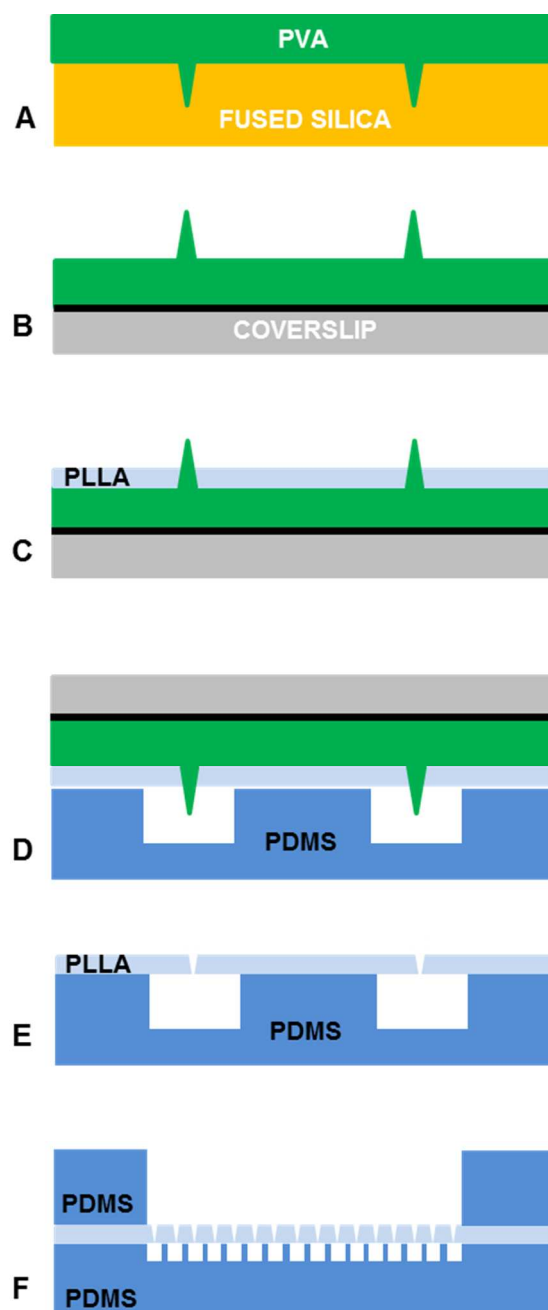
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41 130 Once the femtosecond laser patterned fused silica mold was fabricated, the freestanding
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43 131 polymer nanoneedle arrays were produced using the novel solution-casting mold-replication
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45 132 method recently reported by Rajput and co-workers.^{31,32} In the present work, arrays of
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47 133 freestanding polyvinyl alcohol (PVA) nanoneedles (or PVA replicas) were prepared using the
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49 134 water/alcohol-based PVA mold release agent Partall® Film #10 (Rexco). In particular, each
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51 135 fused silica mold containing an array of surface pores was coated with a thin layer of Partall®
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53 136 Film #10 using a foam paintbrush. The layer was exposed to a flow of nitrogen and allowed to
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3 137 dry (Fig.1a). The resulting ~25 μm thick PVA film with nanoneedles was peeled-off the fused
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5 138 silica mold and mounted on a 22 \times 22 mm^2 glass or polyvinyl chloride (PVC) coverslip using
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8 139 double-sided adhesive tape (Fig. 1b).
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10 **Preparation of PLLA membranes**

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13 141 Spin-coating-assisted deposition of PLLA on the PVA replicas was performed using the
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15 142 same processing parameter values used to prepare plain ~100 nm thin PLLA films reported in
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17 143 previous work³³. A 10 mg/mL solution of PLLA in dichloromethane (molecular weight 100 kDa;
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19 Polysciences, Inc.) was applied to the PVA nanoneedle array and spun at 3000 rpm for 30 s
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22 145 inside a Class 100 clean room (Fig. 1c). The resulting PLLA films were allowed to dry for 1
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24 146 minute at 80 $^{\circ}\text{C}$ and stored in polystyrene Petri dishes.
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147 **Fig. 1-** Process for fabrication of a semipermeable ultrathin PLLA membrane. a) PVA
148 microneedle array is created by casting the PVA solution on a fused-silica mold and by
149 allowing the PVA to dry. b) PVA replica is lift-off bonded to coverslip using adhesive
150 tape (black solid line). c) PVA replica is spin coated with PLLA. d) PLLA film is placed
151 in contact and aligned to the PDMS microfluidic layer. e) PLLA film is adherent on the

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3 152 PDMS layer. f) The final assembled device presents the PLLA barrier between two
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6 153 microfluidic chambers.

8 154 **Assembly of microfluidic devices**

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10 155 A simple two-compartment device was assembled to test the semipermeable PLLA
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12 156 membranes in an organs-on-a-chip assembly. The device consists of two microfabricated layers
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15 157 in polydimethylsiloxane (PDMS, Sylgard® 184 silicone elastomer kit from Dow Corning, MI,
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17 158 USA), produced by SU8 softlithography³⁴. Briefly, the bottom layer consists of 16 parallel
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20 159 channels (200 μm width, 100 μm height, 10 mm length, and separated by a 200 μm space),
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22 160 obtained by casting and curing liquid PDMS (10:1) on SU8-2100 mold (from Microchem, MA,
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25 161 USA) fabricated in a Class 100 clean room. The upper chamber was prepared by punching a
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27 162 thick 3 mm layer of PDMS with a sterile disposable 6 mm diameter punch. Ports to access the
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30 163 microfluidic channels were opened by punching 1.59 mm (1/16 inch) diameter holes.

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32 164 In order to bond a PLLA membrane between the two PDMS layers, the channeled PDMS
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34 165 layer was gently pressed against the PLLA layer formed atop a PVA replica (Fig. 1d). These two
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36 166 pieces were then immersed in de-ionized water for 6 to 12 hours to allow water to gradually
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39 167 dissolve and remove the sacrificial PVA template, leaving behind a semipermeable ultrathin
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41 168 PLLA membrane with precisely patterned micrometer scale pores (Fig. 1e). The resulting PLLA
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44 169 membrane remained adherent to the channeled PDMS layer. The upper PDMS layer chamber
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46 170 and the PLLA membrane were then oxygen-plasma treated (600 mTorr, 100 W, for 45 sec) and
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48 171 finally bonded together (Fig. 1f). Oxygen-plasma treatment renders the exposed surfaces
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51 172 hydrophilic. Once bonded the devices were immediately filled with deionized sterile water and
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53 173 stored at 4°C until used.

55 174 **Cell culture, staining and assays**

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3 175 Human umbilical vein endothelial cells (HUVECS)³⁵ were isolated from umbilical cord
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6 176 obtained from a de-identified placenta collected from patients who underwent elective cesarean
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8 177 sections between 37 and 39 weeks of gestation. All procedures related to the consent and
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10 178 collection of tissues were approved by the Vanderbilt University Institutional Review Board.
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12 179 Cells were cultured in EBM-2 media supplemented with 10% of fetal bovine serum (Lonza,
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14 180 USA) and with 1% antibiotics/antimycotics. Purity of the isolation was validated by
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16 181 morphological and immunofluorescent staining for CD31 (DAKO, USA) before loading in the
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18 182 devices. Cells were maintained at 37°C in a saturated humidity atmosphere containing 95% air /
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20 183 5% CO₂, and they were sub-cultured before reaching 70% confluence (approximately every 2
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22 184 days).
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27 185 After trypsinization and centrifugation, cells were suspended in full medium (2000
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29 186 cells/ μ L) and 50 μ L were injected in the device. Medium was refreshed in the device every 2
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31 187 days. Cell growth on the PLLA nanofilm was monitored for 7 days as proof of principle.
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34 188 Vitality was investigated after 7 days in culture by using NucBlue® Live ReadyProbes®
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36 189 Reagent and NucGreen® Dead 488 ReadyProbes® Reagent (Molecular Probes, R37605 and
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38 190 R37109). 20 μ L of staining solutions were added directly to cells in full media and incubated for
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40 191 20 minutes. NucBlue® Live Cell Stain emits at 460 nm when bound to DNA while NucGreen®
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42 192 Dead 488 reagent is a membrane-impermeable stain DNA of dead cells (excitation/emission at
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44 193 504/523 nm). Cellular cytoskeleton was visualized by F-actin staining using ActinGreen™ 488
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46 194 ReadyProbes® Reagent from Molecular Probes. Cells were prepared for staining by 4%
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49 195 paraformaldehyde fixation.
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3 196 Cells were finally observed with an inverted fluorescent microscope (EVOS, FL Cell
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6 197 Imaging System) and with Image J software (Rasband, W.S., ImageJ, U. S. National Institutes of
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8 198 Health, Bethesda, Maryland, USA).

10 199 **Loading of microfluidic devices**

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13 200 The devices were sterilized by UV irradiation for 30 minutes prior to loading with cell
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15 201 medium and cells. Mimicking the seeding protocol used for cell culture in traditional transwell
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17 202 inserts, the water was replaced with EBM-2 cell-culture media supplemented with 10% of fetal
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19 203 bovine serum (Lonza, USA), first loading the lower microfluidic channels and then the upper
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21 204 chamber. The nanofilm was not coated with any adhesive protein other than that in the media.
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24 205 The wet device was then enclosed in a Petri dish and equilibrated at 37°C for 12 hours inside the
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26 206 incubator.

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31 208 **RESULTS**

33 209 **SEM characterization of fused silica molds and PVA replicas**

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36 210 The surfaces of both the femtosecond laser patterned fused silica mold and the
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38 211 corresponding PVA replica were imaged using a JEOL 6320 field emission scanning electron
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40 212 microscope (SEM). Both samples were sputter-coated with a few nanometers of platinum for
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42 213 successful imaging. An image of a single-pulse femtosecond laser machined pore is shown in
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44 214 Figure 2a. Images of the PVA replica, at various magnifications, are shown in Figures 2b-d.

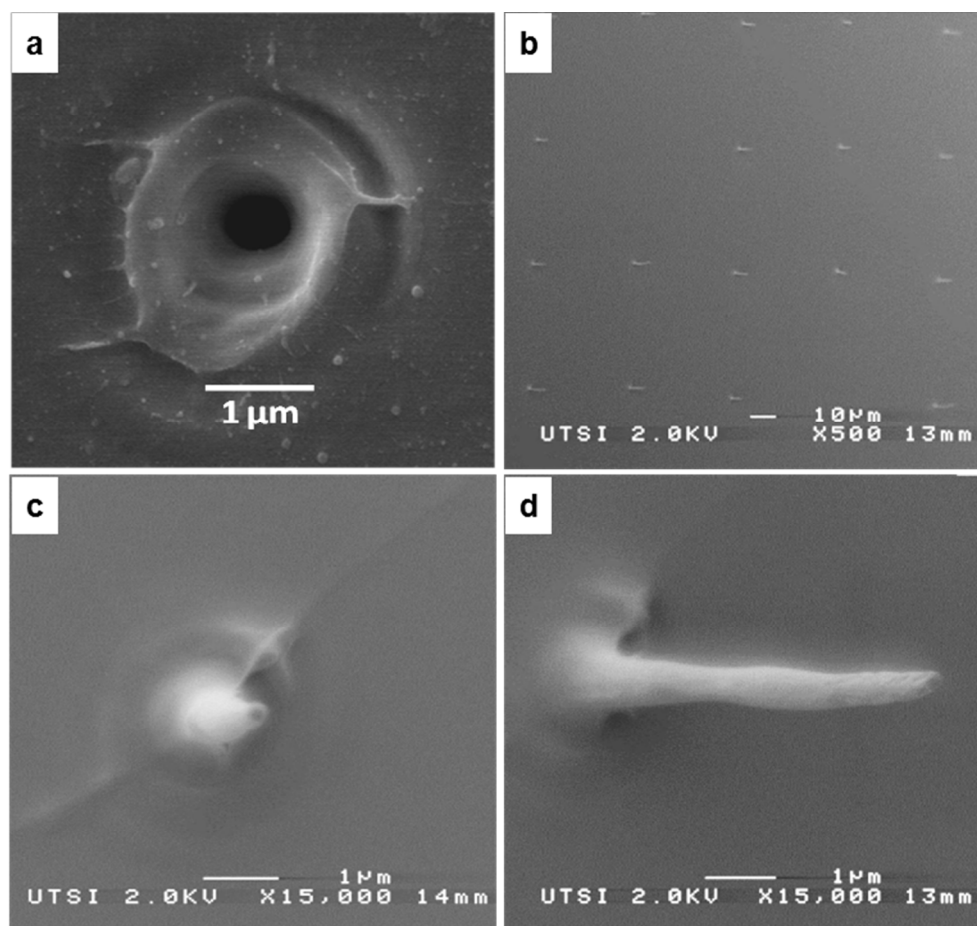


Figure 2 – Process characterization. a) SEM image of a femtosecond laser machined pore on the surface of a fused silica wafer. The pore was opened using a single 3.2 μJ, 790 nm, 160 femtosecond laser pulse; b) SEM image of a PVA replica (Pt coated, 45° stage tilt). c) A single PVA nanoneedle with a final length of 10 μm. d) same PVA nanoneedle at 30° stage tilt.

SEM characterization of PLLA membrane

The effectiveness of the spin coating of PLLA on the PVA microneedle array is apparent in Fig. 3d: the PLLA solution spread and coated the full surface of the replica. Due to the high

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3 225 aspect ratio of the PVA needles and lack of wettability, the tips of the needles were not
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5 226 completely covered during the spin coating (Supporting information, Fig. S1).
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8 227 The sacrificial layer PVA was dissolved by immersion in de-ionized water, and once
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10 228 removed, the PVA needle sites become open pores in the PLLA nanofilm membrane. While in
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12 229 this report, the PLLA film was transfer bonded to the PDMS, the hydrophobic PLLA film can be
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14 230 floated freestanding on the water surface for visualization (Fig. 3a). The patterned pores can be
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16 231 visualized in bright field once the film is collected and dried on a flat surface, such as a silicon
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18 232 wafer or a glass coverslip (Fig. 3b-c). The PLLA nanofilm replicates perfectly the surface
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20 233 morphology and any irregularity of the PVA replica, including the features imposed by the laser
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22 234 in the original silica mold (Fig. 3d).
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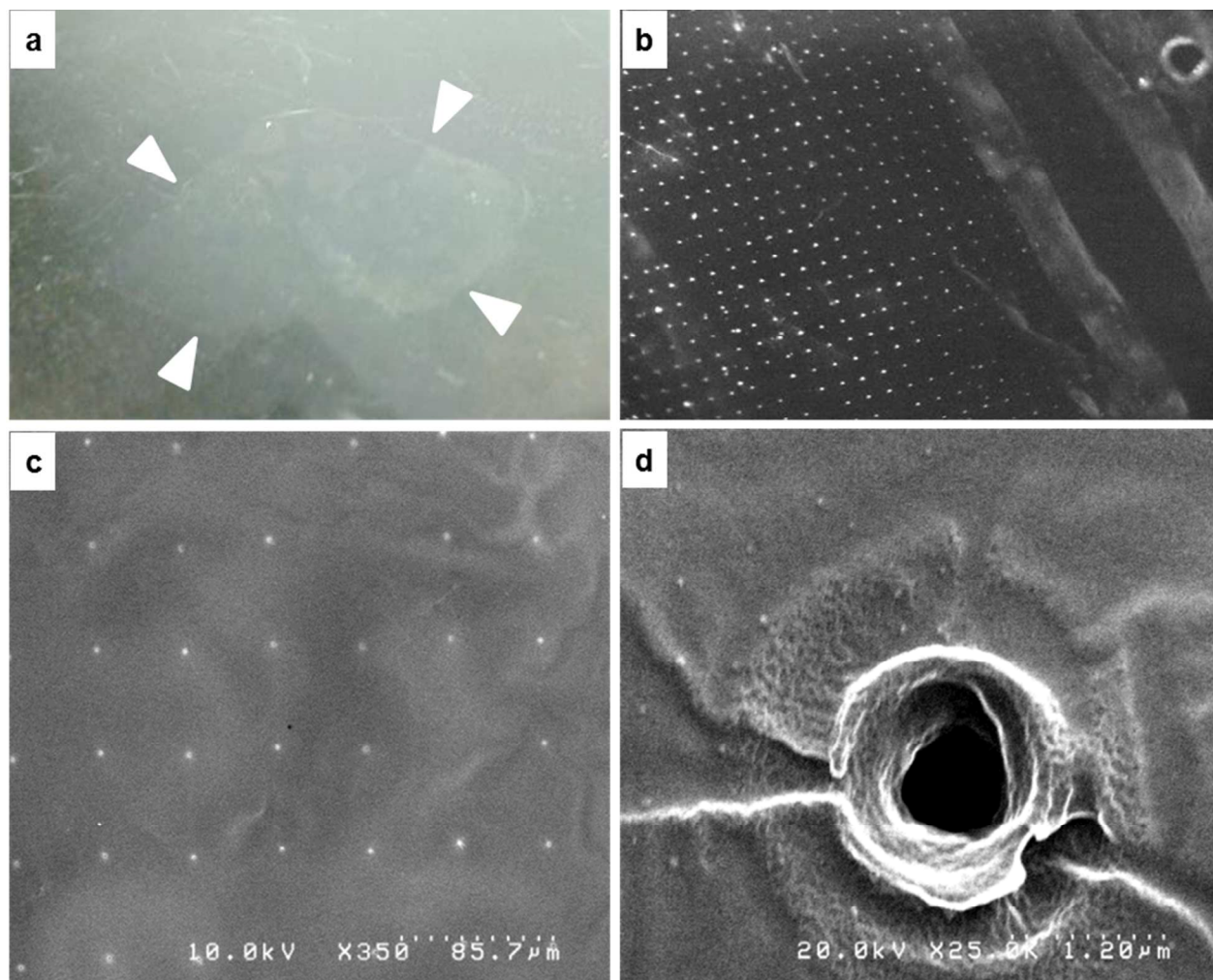


Fig. 3 – Characterization of PLLA film. a) film floating on water surface after the PVA template dissolved (film edges highlighted with white arrows); b) optical microscope image of the film (holes appear as white dots); c) the array of holes in the film are visualized with the SEM (2 nm gold coating); d) A detail of a single hole in the PLLA film.

Different views of a PLLA semipermeable membrane bonded to the channeled, bottom PDMS layer are shown in Fig. 4.

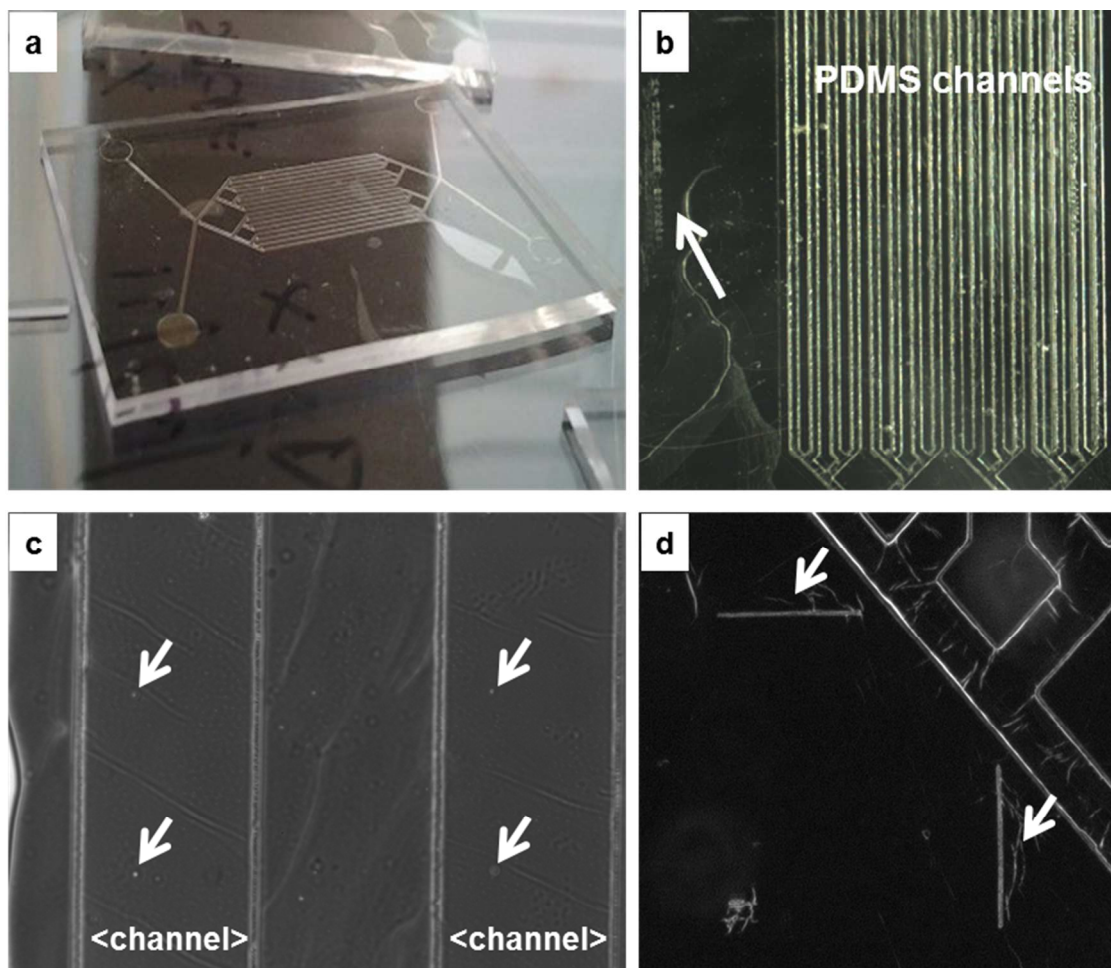


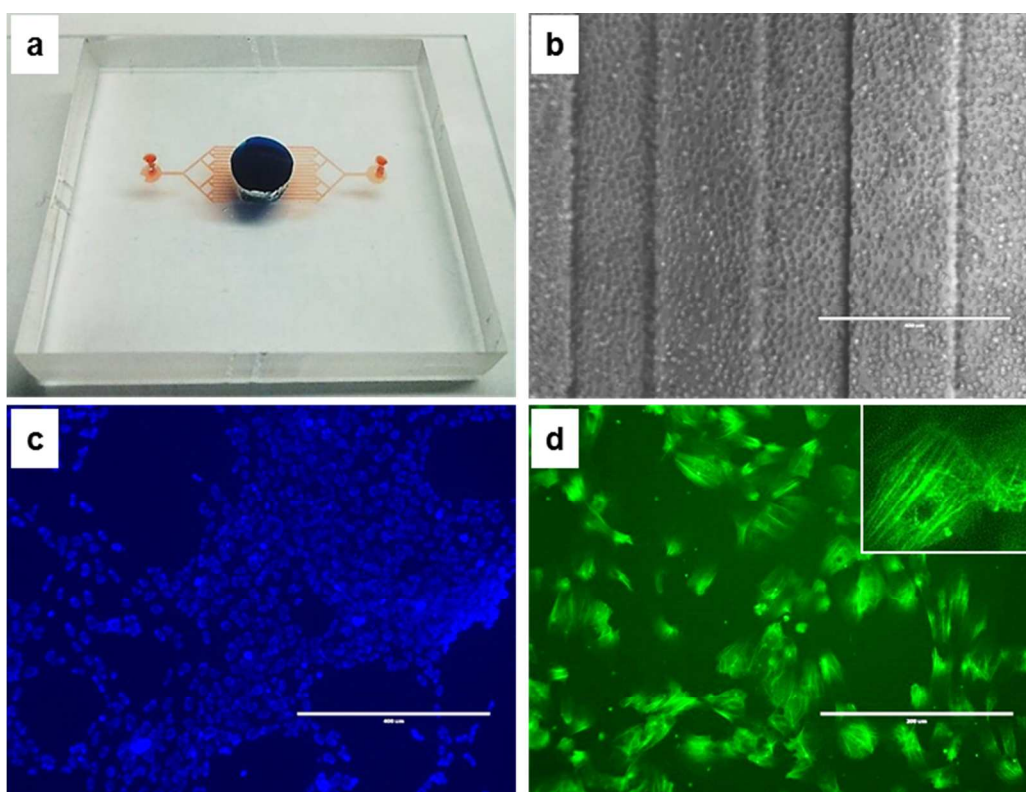
Fig. 4 - Device assembly. a) PLLA film adherent to channeled, bottom PDMS layer; b) PLLA film on a channeled PDMS bottom layer with arrow pointing to the lateral identification label; c) magnified view of Figure 4b showing a group of four micrometric pores in the PLLA membrane laying above the 200 μm-wide microfluidic channels in the PDMS layer; d) alignment features.

Device Assembly

The PLLA membrane bonded wrinkle-free to the channeled PDMS layer (Fig. 4a). The sixteen rows of micrometric pores of the PLLA membrane aligned precisely above the sixteen-microfluidic channels. The sacrificial PVA layer provides a means of handling the nanofilm and

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3 254 mounting the film on the channeled PDMS layer preventing damage to the film; a key concern in
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5 255 manipulation of the ultrathin membrane.

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8 256 Since both PDMS and PLLA are optically transparent, alignment of various components
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10 257 was done under a stereomicroscope. Together with the pattern of surface pores, the original silica
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12 258 mold has femtosecond laser machined surface alignment marks and a lateral identification label
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14 259 (Fig. 4.b-d); these microscopic features transfer to the PVA and PLLA films and serve as
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16 260 alignment guides during device assembly.



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262 **Fig. 5 - Cells adhesion in the device.** a) Full device (blue dye for upper chamber
263 volume, red dye for the lower, channeled chamber volume); b) HUVECs at the time of
264 loading (D0, 4x, ph); c) LIVE/DEAD staining of the cells (green for dead cells, blue for
265 live cells, scale bar 400 μm); d) actin staining of HUVECs inside the device (green for
266 actin, scale bar 400 μm ; inset at 40x).

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3 267 We observed that the PLLA membrane adhered tightly to the channeled PDMS layer and
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5 268 did not collapse inside the channels (Fig. 5a). Cells were then seeded directly on the PLLA
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8 269 membrane in the upper chamber without prior coating of the surface. Approximately 100,000
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10 270 cells were seeded at Day 0 in the device, filling the upper chamber without leakage between the
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12 271 layers. It was possible to observe the cell behavior and monitor growth by optical microscopy
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15 272 due to the transparency of the PLLA membrane in bright field.
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17 273 Oxygen plasma treatment did not adversely affect the biological properties of the PLLA
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19 274 film surfaces; the grow rate and cell morphology of the cells on PLLA inside the device were
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21 275 comparable to the control flask. As shown in Fig. 5c-d, cells proliferate rapidly in the device and
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23 276 with very low mortality (below 1%). The media of the two chambers was replaced every 48
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25 277 hours and after 7 days the adhesion of the cells on the semipermeable PLLA membrane was
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27 278 confirmed by F-actin staining (Fig. 5.d). Cells were trypsinized and the suspensions from the
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29 279 two chambers were collected to measure the amount of cells by using a cell counter. Cells were
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31 280 found only in medium from the upper chamber, thus proving that the size of the pores did not
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33 281 exceed the nominal value of 1 μm in diameter (data not shown). At the same time, the
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35 282 permeability of the membrane was confirmed by 150 kDa FITC-dextran diffusion measurements
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37 283 (as reported in Supporting information, Figure S2).
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46 285 **DISCUSSION AND CONCLUSIONS**

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48 286 Fabrication of biocompatible, transparent, semipermeable ultrathin polymer membranes
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50 287 with precisely patterned micropores is difficult by perforation using e-beam lithography, focused
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52 288 ion beam milling, or laser micromachining. Additionally, such processes lack production
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54 289 scalability. Forming membranes using a molding approach is also difficult to implement. First of
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3 290 all, the polymer melt, polymer precursor, or the polymer-solvent solution being used must have
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5 291 sufficiently low viscosity to spread and fill the surface of the mold completely and
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8 292 homogeneously, without leaving a residual layer on top of the mold micropillars. Mechanical
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10 293 peeling of the membrane is problematic due to the thickness of the membrane. Once released
11
12 294 from the mold, the membrane must be transferred and precisely attached to the microfluidic
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14 295 device and without any wrinkles and folds.

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17 296 The method reported in this paper provides a quick, repeatable and convenient route to
18
19 297 prepare multiple copies of the desired membranes. First, the direct one-step femtosecond laser
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21 298 machining of the fused silica molds can be carried out at kilohertz rates. With the current system,
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23 299 this translates to patterning a regular array of four million surface pores inside a one square
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25 300 centimeter area in less than one hour. Molds can be prepared with a distance between pores as
26
27 301 small as 2 μm . The construction of the silica mold with alignment marks allows the user to
28
29 302 control the size and location of each pore, and more importantly, the density of pores. While in
30
31 303 this demonstration, each microfluidic channel had a single line of 100 pores each separated by
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33 304 100 μm , the ability to separate pores by as little as 2 μm would allow us to have tens of
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35 305 thousands of pores in each channel. The ability to control the location of pores means that the
36
37 306 femtosecond laser drilling need not be wasted on regions of membrane sandwiched between
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39 307 solid PDMS, and the presence of pores and any associated surface irregularities is avoided in
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41 308 such regions where PDMS bonding is critical to avoids leaks and channel cross-talk.

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43 309 Chemically inert and mechanically hard, the fused silica mold can be used to prepare
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45 310 thousands of PVA replicas. The PVA formulation used does not stick to fused silica. Each PVA
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47 311 replica precisely and repeatedly duplicates all the features of the mold, yielding straight,
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49 312 standing, and aligned high aspect-ratio nanoneedles. PVA replicas can be produced relatively
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3 313 quickly (15-30 minutes) and the mold can be easily cleaned from residual traces of PVA by hot
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5 314 water. The water-solubility and organic solvent-resistance of PVA makes it a suitable sacrificial
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8 315 material to use as template in preparing semipermeable membranes for many organic solvent-
9
10 316 soluble polymers of interest. Finally, the PVA template provides a convenient vehicle to transfer,
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12 317 align and attach the ultrathin membrane to the microfluidic device without introducing
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14
15 318 undesirable wrinkles or folds in the final assembly.

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17 319 While the selection of biocompatible polymers is wide and the fabrication of films with
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19 320 micron thicknesses can be achieved with several methods, handling of a perforated thin
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21 321 membrane represents a technological challenge. In this work we selected PLLA since its
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23 322 biocompatibility is well demonstrated. Furthermore, the PVA sacrificial layer is fundamental for
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25 323 handling and positioning of the ultrathin membrane PLLA: while an unpatterned sacrificial layer
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27 324 is traditionally used to release positioning polymeric films from their support, for example in
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29 325 stretchable and epidermal electronics^{33,34}, in this case the dissolvable PVA includes not only the
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31 326 needles but also temporary labels and frames to align the pattern of needles with the
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33 327 microfabricated channels in the PDMS.

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35 328 The semipermeable property of these PLLA membrane guarantees the possibility to
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37 329 separate two fluidic compartments and to select the size of particulates (*i.e.*, cells versus
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39 330 macromolecules) able to pass through the pores. This is a key characteristic in lab-on-a-chip
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41 331 technology, where cellular extravasation and passage of chemical species between
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43 332 communicating compartments need to be controlled. Organs-on-a-chip seeks to understand
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45 333 complex processes; *e.g.*, organ development, embryogenesis, tumor metastasis, and leukocyte
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47 334 infiltration that are regulated by cellular responses to multiple (competing) chemokines as well
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55 335 as autocrine feedback loops, cell-cell interactions, and mechanical stress. The result outlined
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3 336 herein of a scalable, tunable semipermeable nanofilm membrane will contribute to the success of
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5 337 organs-on-a-chip models and enable more faithful recapitulation of *in vivo* conditions.
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10 339 **ASSOCIATED CONTENT**

11
12 340 **Supporting Information.** This section provides additional images of PVA replicas coated with

13 341 PLLA and closer view of the PVA needles after coating. Additionally, the results of the

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15 342 permeability measurements are summarized. This material is available free of charge via the

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17 343 Internet at <http://pubs.acs.org>.

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30
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16 365 Science and Engineering.
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21 367 **ABBREVIATIONS**

22 368 poly (L-lactic acid) (PLLA)

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24 369 polyvinyl-alcohol (PVA)

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26 370 basement membrane (BM)

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28 371 polydimethylsiloxane (PDMS)

29
30 372 human umbilical vein endothelial cells (HUVECS)

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32 373 polyvinyl chloride (PVC)

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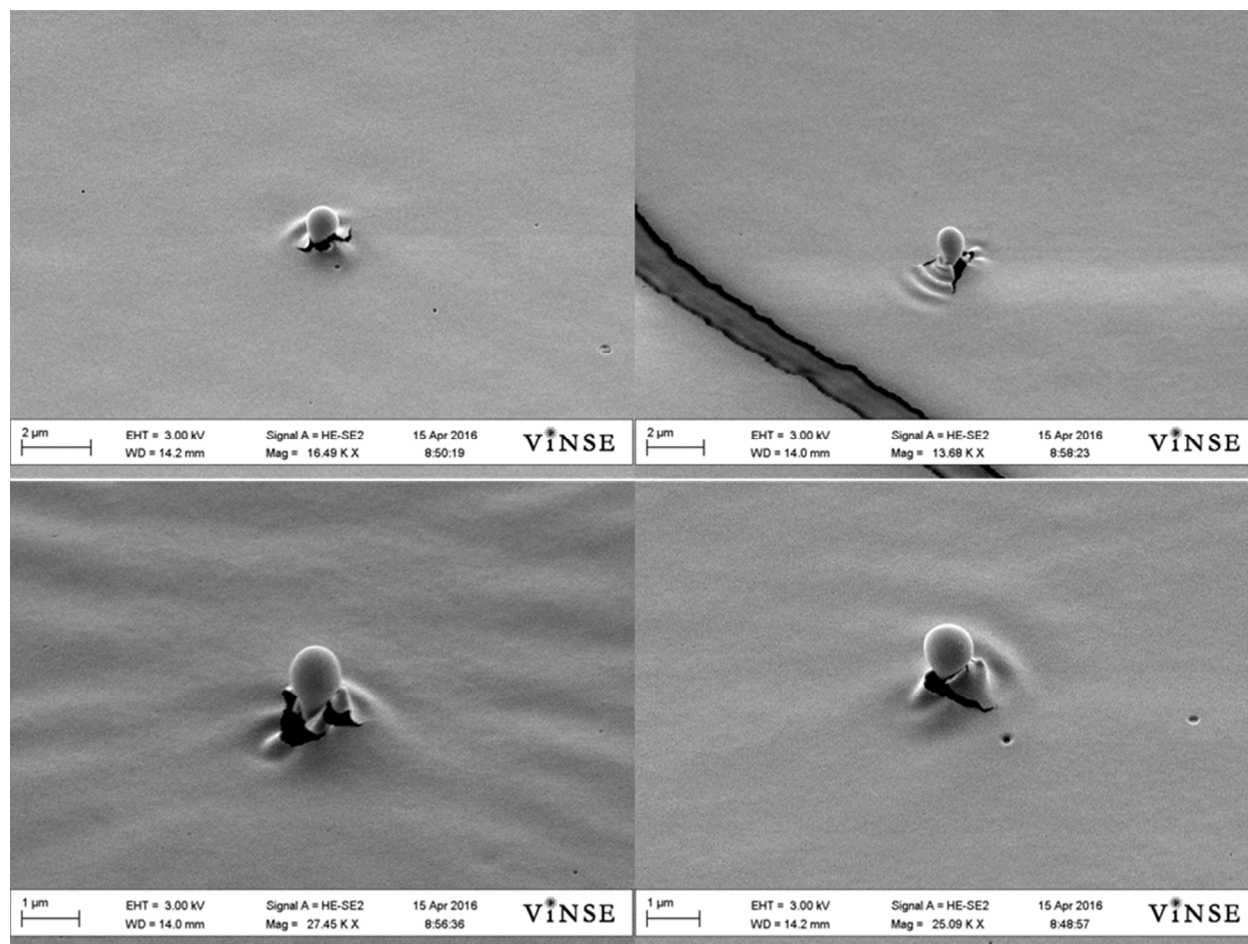
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Supporting Information



484

485 **Figure S1. SEM images of PLLA coated PVA nanowires (Pt coated samples, 30° stage tilt)**

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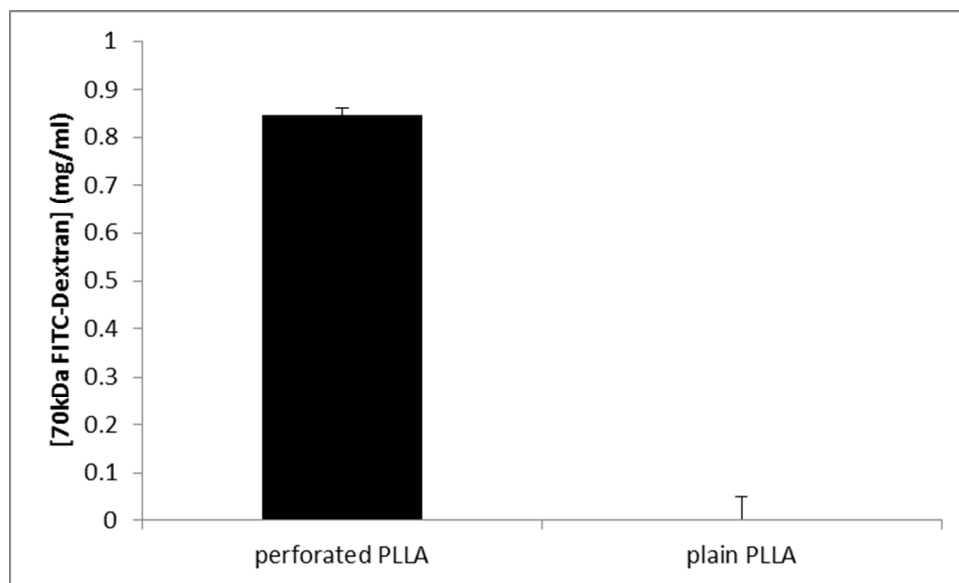
487 **Permeability of the PLLA nanofilms:**

488 Diffusion of FITC dextran (70 kDa molecular weight, Sigma Aldrich, USA) through
489 the perforated PLLA membranes was measured by using a UV-Vis Spectrophotometer
490 (Varian Cary 50, Agilent Technologies, Santa Clara, CA).

491 The experiment was performed in a 96-well plate. Five devices were assembled,
492 following the protocols described in the Materials and Methods section, each one integrating one
493 PLLA perforated membrane. FITC dextran solution (2 mg/mL in MilliQ water) was used to fill

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3 494 the upper chamber of the device, while the lower channels were filled with MilliQ water. Liquid
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6 495 from the lower channels was collected after 6 hours. One additional device was assembled with a
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8 496 PLLA membrane prepared with the same process parameters on a flat silicon wafer, which thus
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10 497 did not have holes and was not permeable.
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36 500 **Figure S2. Permeability of PLLA membranes to FITC-dextran.**

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